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Highlights

- ZnS thin films composed of porous nanoparticles have been deposited.
- Methodology is based on a combination of three techniques.
- Cubic phase ZnS nanoparticles deposited onto glass substrates.
- Films characterized by UV/Vis, PL, XRD, SEM, TEM, AFM and XPS

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Abstract

ZnS thin films composed of porous nanoparticles have been deposited on to glass substrates by combining three simple synthesis methodologies *i.e.* chemical bath deposition, co-precipitation and spin coating. The XRD results reveal the cubic phase of ZnS thin films crystallized at nano scale. The crystallite size estimated by Scherrer formula was 3.4 nm. The morphology of the samples was analyzed through scanning electron microscopy (SEM) and is evident that thin films are composed of porous nanoparticles with an average size of 150 nm and pores of 40 nm on almost every grain. Crystallinity, phase and morphology were further confirmed *via* transmission electron microscopy (TEM). The stoichiometry and phase purity of thin films were determined by energy dispersive X-ray (EDX) spectrum and X-ray photoelectron spectroscopy (XPS) analysis, respectively. The surface topography and homogeneity of thin films were analyzed by atomic force microscopy (AFM) and obtained root mean square roughness (4.0326 nm) reveals the morphologically homogeneous growth of ZnS on glass substrates. The UV-Vis spectroscopy and photoluminescence (PL) were carried out to estimate the band gap and observe the emission spectra in order to speculate the viability of ZnS porous nanoparticles in optoelectronic devices and sensors.

Keywords: Thin films, semiconductors, chemical synthesis, precipitation, electron microscopy

1. Introduction

In the recent years, extensive studies have been carried out on the synthesis/deposition and characterization of large band gap semiconductors due to their applications in photovoltaic, photo-electrochemical energy conversion and photoconductors [1, 2]. ZnS is an important II-VI semiconductor with large band gap and have been studied extensively regarding its viability in above mentioned applications. Porous nanoparticles, thin films and structures have attained focus of research since they exhibit novel properties and have many applications in photo-catalytic, luminescent and sensing devices [3-5]. Several methods have been employed to synthesize ZnS thin films including: chemical spray pyrolysis [6, 7], solvothermal synthesis [8, 9], sol-gel [10], successive ionic layer adsorption and reaction (SILAR) [11], pulsed laser deposition (PLD) [12], metal-organic chemical vapor deposition (MOCVD) [13], RF-magnetron sputtering [14, 15], electrodeposition [16, 17], thermal evaporation [18], electron beam evaporation [19], aerosol assisted chemical vapour deposition (AACVD) [20, 21] and chemical bath deposition (CBD) [22-30]. Amongst all, CBD is a simple and cost effective method to deposit high quality thin films at relatively low temperatures without need of ultra-high vacuum and sophisticated instrumentation. Synthesis of un-doped and doped ZnS nanoparticles by various methods has also been reported previously. Most recently, Zhu *et al.* [31] reported the sonochemical synthesis of mesoporous ZnS nanocrystals without using template/surfactant. The prepared nanomaterials exhibit higher photo-degradation activity of Rh under UV irradiation than that prepared in absence of sonochemistry or in an aqueous system. Pathak *et al.* [32] reported the synthesis of un-doped and cobalt doped ZnS nanoparticles by co-precipitation method. However literature suggests that the study and growth of porous ZnS nanoparticles is lacking. To the best of our knowledge, the deposition of porous ZnS nanoparticles on glass substrates without use of

surfactants or templates by any of the chemical technique or combination of them has not been reported yet. However, few reports have been published on study of luminescent properties of ZnS thin films deposited on porous substrates (Si) by different techniques [33].

We have reported the deposition of uniform ZnS thin films by an optimized CBD method [34]. Herein, we report a novel approach to deposit thin films composed of porous ZnS nanoparticles by simple CBD/co-precipitation and spin coating method. The observed green PL emission of the ZnS nanoparticles and their porosity are important factors for the development of novel luminescent devices and sensors.

2. Experimental

2.1 Chemicals

All reagents, zinc chloride, thioacetamide, urea and hydrochloric acid were purchased from Sigma-Aldrich and used without further purification. The de-ionized water was used as solvent. Acetone and isopropyl alcohol (IPA) were used for cleaning of the glass substrate. Glass substrates have already been established as the best substrates to study the optical properties of the thin films as compared to sapphire or MgF_2 substrates due to their negligible interaction on the optical properties of the material deposited onto them. Analytical grade ethanol was used for spin coating the films.

2.2 Instruments

X-Ray diffraction measurements were performed using Bruker D8 advance Diffractometer with Cu-K α radiation. Data were recorded across a 2θ range of 20 to 80° with a step size of 0.05°. SEM and EDX analyses were carried out using Philips XL 30 microscope. TEM, HRTEM and SAED images were collected from Tecnai 20 F30 transmission electron microscope using

accelerating voltage of 200 kV. An atomic force microscope (AFM) PeakForce QNM was used to measure surface roughness of the ZnS thin films. Absorbance and transmittance spectrum was acquired using Agilent HP 8453 UV-Vis spectrophotometer. Fluorolog 22 HORIBA JOBINYVON was used to obtain photoluminescence data using excitation wavelength of 340 nm.

2.3 Synthesis of porous ZnS thin films

The synthesis of porous ZnS thin films was carried out in three steps by using CBD, co-precipitation and spin coating, separately.

2.3.1 CBD/co-precipitation

Chemical bath deposition/co-precipitation of ZnS was carried out first. Chemical bath containing zinc chloride (0.2 M), urea (3 M) and thioacetamide (0.4 M) was used to synthesize ZnS. The solutions of zinc chloride (40 mL), thioacetamide (40 mL), and urea (20 mL) were mixed in a beaker to get total volume of 100 mL bath solution. The pH of bath solution was adjusted to 4.0 by drop wise addition of 1.0 M HCl. The stirred bath solution was maintained at a temperature of 80 °C. The reaction process for the formation of ZnS in CBD is based on, first the slow release of zinc and sulfur ions within the solution followed by condensation of these ions [30]. Balance between hydrolysis and condensation can be adjusted by the presence of urea in the solution [35]. Conventionally, in CBD, the deposition of thin films would be carried out under uniform magnetic stirring to avoid precipitation throughout the reaction time. After the deposition of thin films, the reaction mixture is usually disposed. In the present experiment, the mixture was stirred for 3 hours at 80 °C and then the stirring was stopped while the mixture was further heated for 1 hr. The heating was stopped and the white precipitate formed was

centrifuged and washed three times with methanol to remove the byproducts of reaction. The precipitate was then dried under vacuum at room temperature and kept under nitrogen. The powder obtained after drying was then used to deposit the thin films of ZnS on glass substrates by spin coating.

2.3.2 Spin coating of ZnS thin films

The dried powder was aged at room temperature under nitrogen stream for 24 hours to avoid oxidation. For spin coating, powder samples were dispersed in ethanol and were deposited on pre-cleaned glass substrates. The solution was spun at 500 rpm for first 5 sec. and then at 3000 rpm for further 25 sec. in order to achieve uniform coating of material onto the substrates. Three subsequent layers were deposited to achieve workable thickness of the samples. After deposition, all the samples were annealed under nitrogen atmosphere for one hour in tube furnace at 400 °C to remove any volatile by-products and improve the crystallinity before further characterization.

3. Results and Discussion

3.1 Structural studies

Figure 1 shows the typical X-ray diffraction pattern of ZnS films. The diffraction peaks are intense and broad, suggesting the formation of crystallites with size in the nano regime. The major peaks of pure ZnS are observed. Three broad peaks corresponding to the (111), (220) and (311) lattice planes are well matched with reported data (ICSD # 01-080-0020) and can be assigned to the cubic phase of ZnS. The vertical lines on x-axis represent the standard pattern of cubic ZnS. The XRD pattern shows that the thin films are single phase since there is no diffraction peak regarding any impurity or other (hexagonal) phase of ZnS. A small but bit broader peak (hump) observed between the 2θ values of 20-25° is due to glass substrate.

Fig. 1: XRD pattern of ZnS thin film

From the full width at half maximum (FWHM) of most intense peak (111), the average crystallite size was estimated by Scherrer's equation as; $D = 0.9\lambda / \beta \cos\theta$, where D is the crystallite size, λ is the X-ray wavelength of Cu-K α radiations ($1.5405\text{\AA}$), β is the full width at half maximum (in radian), and θ is the Bragg diffraction angle. The calculated crystallite size was 3.4 nm. Cell parameters calculated by the relation; $a = d(\sqrt{h^2 + k^2 + l^2})$ and found as $a = b = c = 5.32\text{\AA}$ which are close enough to the reported values of lattice constants in ICSD # 01-080-0020. The dislocation density (δ), defined as the length of dislocation lines per unit volume was estimated using the equation [36]; $\delta = \frac{1}{D^2}$ and obtained as 0.0865 nm^{-2} . Lattice strain (ϵ) in the thin film was estimated using the equation [37]; $\epsilon = \frac{B\cos\theta}{4}$ and obtained a value of -0.193 showing that the material was deposited with least value of strain in the lattice.

3.2 Morphology and Stoichiometry

The SEM micrographs at different magnifications for the ZnS thin film are shown in Figure 2. Images at high magnifications (Figure 2 (a-b)) reveal that films are composed of porous nanoparticles. Low magnification image (Figure 2c) reveals that deposit is in the form of almost spherical agglomerates of particles with their size in nanometer range and is deposited uniformly throughout the substrate surface (Figure 2d). The apparent size of most of the particles is less than 200 nm and the size of pores on almost every grain is 40 nm (Figure 2a). Thin films of ZnS obtained using same materials by CBD on glass substrate show non-porous and almost spherical nanoparticles [34].

Fig. 2: SEM micrographs (at various magnifications) of spin coated porous ZnS thin film

The spin coating of the residue from CBD might have the major role in obtaining the porous nanoparticles. To validate this argument, SEM analysis of nanoparticles before spin coating was also performed and found that nanoparticles exhibit almost spherical morphology as obtained after spin coating but without porous structure (Figure 3). For that reason, this observation needs further studies to elaborate the in-situ mechanism of spin coated ZnS thin films comprised of porous nanoparticles. Figure 4 shows the particle size distribution in spin coated ZnS thin film evaluated by ImageJ tool. The majority of particles have their size distribution in the range from 120-180 nm.

Fig. 3: SEM micrograph of ZnS nanoparticles before spin coating

Fig. 4: particle size distribution of porous ZnS thin film

Stoichiometry of the elemental composition of ZnS thin films was analyzed through EDX which confirmed that the deposited thin films have Zn:S ratio of ~1:1. Glass constituents such as, silicon, sodium, calcium, magnesium, potassium and aluminum were also detected in EDX spectrum because of the thin nature of the films. Figure 5 shows the EDX pattern of ZnS thin films on glass substrates.

Fig. 5: EDX pattern of ZnS thin films

To support the evidence of ZnS deposition on glass substrate by EDX, X-ray photoelectron spectroscopy was carried out for thin films. Besides the wide scan spectrum in the range of 0-1200 eV, detailed spectra for the Zn_{2p} and S_{2p} regions were obtained and presented in Figure 6. The black and red lines represent the obtained and fitted data, respectively. The XPS spectra of ZnS thin film reveal the formation of stoichiometric and impurity free zinc sulfide. The binding

energies of zinc and sulfur corresponding to their 2p orbitals observed at 1019 and 160 eV respectively. A small shift in binding energies can be attributed to the chemical effects along with matrix effects *i.e.* relaxation energy, work function and crystal potential [38].

Fig. 6: XPS spectra of ZnS thin films deposited by CBD method

The morphology of the deposited porous ZnS nanoparticles was further investigated by transmission electron microscopy. Figure 7 shows the representative transmission electron microscopy (TEM), high resolution transmission electron microscopy (HRTEM) images and selected area electron diffraction (SAED) pattern of ZnS nanoparticles. The TEM observations show that it is difficult to observe the actual size of nanoparticles due to aggregation of particles as seen in the Figure 7 (a). The small crystallite aggregate into secondary nanoparticles since they have extremely small dimensions with higher surface energy due to high surface-to-volume ratio. The observation of monodispersed nanoparticles in TEM samples may be possible by diluting and sonicating the samples for a long time.. HRTEM image (Figure 7b) shows that ZnS nanoparticles are of good crystallinity. Figure 7 (d,e) shows lattice spacing as 0.30 nm corresponding to (111) plane of cubic phase consistent with the results obtained from XRD

analysis. Figure 7 (c) shows the SAED pattern which is analogous to the cubic zinc blende structure of ZnS as investigated by XRD. Polycrystalline nature of ZnS nanoparticles was confirmed by appearance of wide diffraction rings made up of many diffraction spots. The diffraction rings in SAED pattern correspond to (111), (220) and (311) planes of cubic phase of ZnS. The electron diffraction reveals that each grain is composed of many crystallites with extremely small nuclei.

Fig. 7: TEM, HRTEM and SAED images ZnS porous nanoparticles

The surface topography images were obtained from Atomic Force Microscope (AFM) used as a source of information about the root mean square roughness, height of the deposited layer and the size of particular features as grains or pores. Figures 8 and 9 show the AFM data collected for ZnS thin films in non-contact mode. From the 2D/topography image and contour plot of topography (Figure 8(a, b)), it can be observed that grains are uniformly distributed throughout

the substrate surface. 3D image (Figure 8c) of the sample also reveals the distribution of grains and surface morphology. Figure 9 shows the grain size estimation (line profile), topography for root mean square roughness (RMS) and bearing surface/volume plots for CBD deposited ZnS thin films.

In order to determine the appropriate relation between the synthesis methodology and morphology of the surface, the size, shape, orientation and density of grains is very important. Figure 9 (a) shows the line profile to estimate the average grain size ~ 200 nm. The shape and density of these grains along with thickness of the film (~ 400 nm) are in agreement to the SEM micrographs. The surface profile of the films was used to obtain probability density function as Bearing Area Curve (BAC) and Bearing Volume Curve (BVC) shown in Figure 9 (c, d). The root mean square roughness (4.0326 nm) reveals the morphologically homogeneous growth of ZnS.

Fig. 8: AFM images (a) 2D topography, (b) contour plot and (c) 3D surface topography of CBD deposited ZnS thin film

Fig. 9: AFM data analysis images, (a) grain size estimation, (b) topography for RMS and (c, d) bearing surface and volume plots for CBD deposited ZnS thin film

3.3 Optical studies

The absorbance of ZnS thin films was obtained from UV-Vis spectrometer. The transmittance and absorbance plots (figure 10) show 75-95% transmittance in the major portion of visible region and almost 100% transmission in infrared region of the solar spectrum. The high transparency of deposited ZnS thin films would have applications in optoelectronic devices.

Fig. 10: Absorbance and Transmittance of CBD deposited ZnS thin films

Absorption co-efficient was calculated and is related to the optical band gap by the relation $\alpha = \left(\frac{k}{h\nu}\right)(h\nu - E_g)^n$, Where k is a constant, h is Planck's constant, $h\nu$ is the incident photon energy and n is a number which characterizes the nature of electronic transitions between valance and conduction bands [39]. For direct allowed transitions $n = 1/2$ and it is known that ZnS is a direct band gap semiconductor. Therefore the formula used is $\left(\frac{k}{h\nu}\right)(h\nu - E_g)^{\frac{1}{2}}$.

Figure 11 shows the variation of $(\alpha h\nu)^2$ with photon energy $h\nu$. The deposited material exhibit direct type of transition as observed from the linear portion of $(\alpha h\nu)^2$ Vs $h\nu$ curve over a wide range of photon energies. The linear portion of the curve was extrapolated to evaluate the energy gap E_g of the ZnS thin films. The intercept on energy-axis estimates the band gap energy (3.70 eV) bit higher than that of bulk cubic ZnS (3.54 eV) [40] due to nanocrystalline nature of thin films [41]. Optical results are completely in co-relation with the XRD analysis since few of the reports determined the hexagonal phase [42, 43] of ZnS by CBD method.

Fig. 11: Band gap plot of ZnS thin film

The photoluminescence spectra for ZnS thin films were carried out at room temperature with an excitation wavelength of 340 nm. Emission spectrum of the nano porous ZnS thin film is shown in Figure 12. Multi-peak Gaussian fitting gives two bands one at 396 nm and the second at 507 nm. The emission peak at 396 nm is related to the sulfur vacancies at the surface of the sample as observed previously [44]. The broad emission band at 507 nm is might be attributed to the electron recombination from sulfur vacancies to the surface defects of ZnS nanoparticles [45]. It is obvious that, growth of thin films composed of nanoparticles with different sizes extensively affect the photoluminescence of the material. The variation in optical properties strongly depends on the existence and density of defects incorporated in the thin films which act as photo-excited carriers and centers of radioactive recombination [46, 47]

Fig. 12: PL of deposited ZnS thin film

The observed luminescent intensity in the present study is many fold higher than that observed earlier by Yang *et al.* [5] in addition to the emission of photons with comparable energy. The porous structure of ZnS nanoparticles could improve intensity of luminescence due to its greater surface area and the fact that existence of pores would enhance the absorption of light. Porous structures also boost the recombination of photo-excited carriers through surface vacancies [48]. The argument presented regarding improvement in properties of materials owing to their porous structure is in strong agreement with the observations reported by Yang *et al.* [49] and An *et al.* [50] for ZnS and ZnO, respectively.

Conclusions

ZnS thin film composed of porous nanoparticles has been successfully deposited onto glass substrate *via* spin coating by using ZnS precipitates, as starting material, collected from CBD/co-precipitation method. The XRD reveals that thin films are nanocrystalline and monophasic with cubic zinc blende phase. Morphological analysis *via* SEM confirms the depositions of porous nanoparticles with pore diameter of 40 nm. The stoichiometric analyses (EDX, XPS) reveal the deposition of impurity free ZnS thin films having stoichiometric ratios of constituents as expected. The UV-Vis results are consistent with the XRD data presenting the deposition of ZnS thin films composed of nanoparticles. In conclusion, nano porous ZnS thin films synthesized by novel three step methodology showed high crystallinity and have potential applications in advanced optoelectronic devices.

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